

## Liquidus Diagram for the Sodium Orthophosphate-Lead Orthophosphate System\*

R. K. OSTERHELD AND J. D. HAWTHORNE

*Department of Chemistry, University of Montana, Missoula, Montana 59801*

Received April 18, 1972

The  $\text{Na}_3\text{PO}_4\text{-Pb}_3(\text{PO}_4)_2$  system was studied by thermal analysis, high-temperature microscopy, and X-ray diffraction powder methods. The compound  $\text{NaPbPO}_4$  melts congruently at  $1117^\circ\text{C}$ . An unidentified phase occurs in the vicinity of 10 mole%  $\text{Pb}_3(\text{PO}_4)_2$ . The system has eutectics at 44.5 and 96 mole%  $\text{Pb}_3(\text{PO}_4)_2$ , melting at 1074 and  $1004^\circ\text{C}$ , respectively. The respective melting points of  $\text{Na}_3\text{PO}_4$  and  $\text{Pb}_3(\text{PO}_4)_2$  are 1512 and  $1015^\circ\text{C}$ .

### Introduction

A variety of compounds of the general formula  $\text{A}^I\text{B}^{II}\text{PO}_4$  have been reported. In the known examples the univalent cations include  $\text{Li}^+$ ,  $\text{Na}^+$ ,  $\text{K}^+$ , and  $\text{Rb}^+$ , and the divalent include  $\text{Mg}^{+2}$ ,  $\text{Ca}^{+2}$ ,  $\text{Sr}^{+2}$ ,  $\text{Ba}^{+2}$ ,  $\text{Mn}^{+2}$ ,  $\text{Co}^{+2}$ ,  $\text{Ni}^{+2}$ ,  $\text{Cu}^{+2}$ ,  $\text{Zn}^{+2}$ ,  $\text{Cd}^{+2}$ , and  $\text{Pb}^{+2}$ . Interest in these compounds stems from luminescent properties and from crystallographic relationships to  $\text{A}^I\text{B}^{II}\text{SiO}_4$  and  $\text{A}^I\text{B}^I\text{SO}_4$  compounds.

The  $\text{A}^I\text{B}^{II}\text{PO}_4$  compounds occur in the  $\text{A}_2^I\text{O}-2\text{A}^I\text{PO}_3\text{-B}^{II}(\text{PO}_3)_2\text{-B}^{II}\text{O}$  quaternary systems at the intersection of the  $\text{A}_3^I\text{PO}_4\text{-B}^{II}(\text{PO}_4)_2$  and the  $\text{A}^I\text{PO}_3\text{-B}^{II}\text{O}$  sections. Phase diagrams have been determined for surprisingly few of these sections. Of the latter type, the  $\text{NaPO}_3\text{-PbO}$ ,  $\text{NaPO}_3\text{-MnO}$ , and  $\text{NaPO}_3\text{-CdO}$  systems have been studied (1), as well as the  $\text{NaPO}_3\text{-MgO}$  and  $\text{NaPO}_3\text{-ZnO}$  systems (2). Of the mixed orthophosphate systems, only part of one has been reported,  $\text{NaCaPO}_4\text{-Ca}_3(\text{PO}_4)_2$  (3, 4). The 25 and  $50^\circ\text{C}$  isotherms of the  $\text{Na}_3\text{PO}_4\text{-Cu}_3(\text{PO}_4)_2\text{-H}_2\text{O}$  system show the occurrence of  $\text{NaCuPO}_4$  (5).

### Experimental

Because of uncertainties in the composition of commercial trisodium phosphate hydrates (6, 7),  $\text{Na}_3\text{PO}_4$  was prepared from  $\text{Na}_2\text{CO}_3$  and  $\text{Na}_2\text{HPO}_4$  (8). Appropriate mixtures of  $\text{Na}_3\text{PO}_4$

and reagent grade  $\text{Pb}_3(\text{PO}_4)_2$  were ground, heated at  $1200^\circ\text{C}$  for 3-4 min, cooled, reground, and annealed at  $800^\circ\text{C}$  for 4 hr. Heating at  $1200^\circ\text{C}$  provided a liquid phase in almost all the samples to increase homogeneity. Possible shifts of sample composition through volatility prevented longer heating or the use of a higher temperature. Crystallinity was improved by annealing at  $800^\circ\text{C}$ .

From 30 to 100%  $\text{Pb}_3(\text{PO}_4)_2$  phase changes were determined by thermal analysis and by high-temperature microscopy (9). Only the hot stage was used at lower lead phosphate levels because of the temperatures involved. Calibration points were  $\text{K}_2\text{SO}_4$  tr.  $583^\circ\text{C}$ , m.  $1069^\circ\text{C}$ ;  $\text{NaCl}$  m.  $800^\circ\text{C}$ ; and  $\text{CaF}_2$  m.  $1330^\circ\text{C}$ . Thermal analyses were carried out at  $10^\circ\text{C}$  per min on 0.9-g samples using a bare 95Pt5Rh-80Pt20Rh thermocouple in the sample. In almost every case the cooling curve and a second heating curve were determined in addition to the original heating curve. The reported temperatures below  $1300^\circ\text{C}$  are felt to be  $\pm 5^\circ$ ; those above  $1300^\circ\text{C}$  have larger uncertainties due to composition changes resulting from partial volatilization of the sample.

### Results and Discussion

X-ray diffraction powder patterns for successive samples across the system showed the occurrence of four phases. The pattern for the

\* This work was supported in part by a grant from the Research Corporation.

first phase,  $\text{Na}_3\text{PO}_4$ , was sharply reduced in intensity at 3%  $\text{Pb}_3(\text{PO}_4)_2$  and undetectable at  $\geq 5\%$   $\text{Pb}_3(\text{PO}_4)_2$ . The pattern for the second phase peaked in intensity at about 10%  $\text{Pb}_3(\text{PO}_4)_2$  and vanished beyond 50%  $\text{Pb}_3(\text{PO}_4)_2$ . With principal lines at the following  $d$  values ( $I/I_0$ ), it could not be matched with any pattern in the ASTM file or otherwise available to us: 3.97 (80), 2.84 (100), 2.78 (90), 2.28 (30), 1.99 (45). The pattern for the third phase showed maximum intensity at about 50%  $\text{Pb}_3(\text{PO}_4)_2$  and was attributed to  $\text{NaPbPO}_4$ . The final phase was  $\text{Pb}_3(\text{PO}_4)_2$ ; its pattern was detectable only at  $>50\%$  of that component.

Temperatures at which phase changes were observed are shown in Fig. 1 for each composition studied. For the compounds  $\text{Na}_3\text{PO}_4$ ,  $\text{NaPbPO}_4$ , and  $\text{Pb}_3(\text{PO}_4)_2$  the respective melting points are 1512, 1117, and 1015°C. Melting point values of 1014 (10) and 1015°C (11) have previously been reported for  $\text{Pb}_3(\text{PO}_4)_2$ . A transition was observed at 340°C for  $\text{Na}_3\text{PO}_4$ .

The  $\text{NaPbPO}_4$ - $\text{Pb}_3(\text{PO}_4)_2$  eutectic occurs at 1004°C and 96%  $\text{Pb}_3(\text{PO}_4)_2$ . A reversible transition at 800°C was attributed to  $\text{NaPbPO}_4$  on the basis of the dependence on composition of the strength of the accompanying thermal effect. Although such a transition is typical of  $\text{A}^1\text{B}^{\text{II}}\text{PO}_4$  compounds,  $\text{NaPbPO}_4$  has previously been reported to be an exception, showing only what should be the high-temperature form (12). It has been shown, however, that the high-temperature

phase of an  $\text{A}^1\text{B}^{\text{II}}\text{PO}_4$  compound can be stabilized at ordinary temperatures by the presence of a substance insoluble in the low-temperature phase, such as sodium carbonate (13, 14).

Because of the high temperatures involved, thermal studies in the composition range 0–20%  $\text{Pb}_3(\text{PO}_4)_2$  were limited to hot-stage microscopy. Fogging of the hot stage cover slide and successive decreases in the liquidus temperature on reheating of samples gave evidence of serious composition shifts in the 3–10%  $\text{Pb}_3(\text{PO}_4)_2$  range. For this range Fig. 1 shows the highest liquidus temperature determined on the first heating of several samples at each composition. It is probable that these temperatures are low. If, as suggested by the X-ray data, a compound exists in this composition range, it is probably not congruent melting; the last crystals to disappear during heating on the hot stage appeared to be  $\text{Na}_3\text{PO}_4$ . It is possible that this apparent compound is a ternary solid solution, comparable to the " $\gamma$ - $\text{Zn}_3(\text{PO}_4)_2$ " observed in the  $\text{Zn}_3(\text{PO}_4)_2$ - $\text{Mn}_3(\text{PO}_4)_2$  system (15). The eutectic between this unidentified phase and  $\text{NaPbPO}_4$  occurs at 1074°C and 44.5%  $\text{Pb}_3(\text{PO}_4)_2$ .

We did not find in this system the previously reported apatite-like compound,  $\text{NaPb}_4(\text{PO}_4)_3$  (16), nor a compound analogous to  $\text{Na}_2\text{Ca}_5(\text{PO}_4)_4$  (17). The former compound forms only in the presence of water vapor (12). Since these compounds would occur at 80 and 71 mole%  $\text{Pb}_3(\text{PO}_4)_2$ , respectively, and the liquidus shape

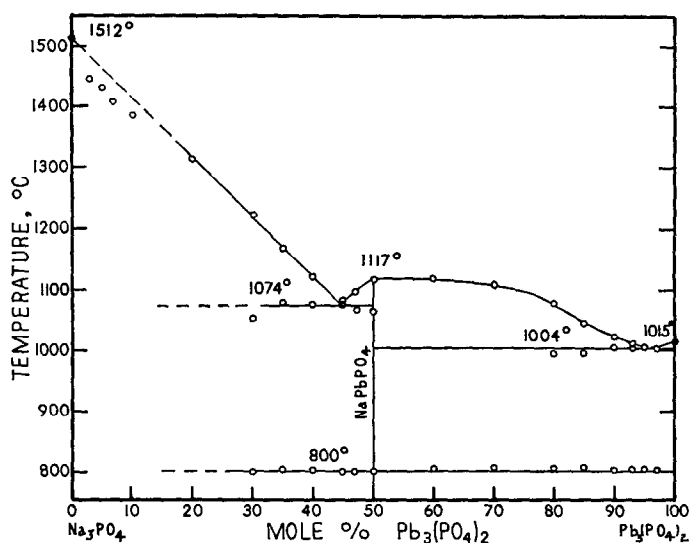


FIG. 1.  $\text{Na}_3\text{PO}_4$ - $\text{Pb}_3(\text{PO}_4)_2$  system.

is unusual in this region, the samples at 80 and 70%  $\text{Pb}_3(\text{PO}_4)_2$  were additionally heated at 800°C for twelve 22-hr periods, the samples being ground between the heating periods. This extended annealing caused no change in the thermal analyses nor in the X-ray diffraction patterns for these samples. The liquidus shape in this region is attributable, presumably, to quaternary features outside this binary section.

## References

1. F. DECARLI, *Atti II Congr. Naz. Chim. Pura Appl.* **1926**, 1146; *Chem. Abs.* **22**, 2119 (1928).
2. S. I. BERUL AND N. K. VOSKRESENSKAYA, *Izv. Akad. Nauk SSSR Neorg. Mater.* **4**, 2129 (1968).
3. J. ANDO AND S. MATSUNO, *Bull. Chem. Soc. Japan* **41**, 342 (1968).
4. S. MATSUNO, T. MIYAHASHI, AND J. ANDO, *Kogyo Kagaku Zasshi* **70**, 1638 (1967).
5. I. G. DRUZHININ AND L. A. TUSHEVA, *Izv. Vyssh. Ucheb. Zaved. Khim. Khim. Tekhnol.* **10**, 1075 (1967).
6. R. N. BELL, *Ind. Eng. Chem.* **41**, 2901 (1949).
7. B. WENDROW AND K. A. KOBE, *Ind. Eng. Chem.* **44**, 1439 (1952).
8. R. K. OSTERHELD AND E. W. BAHR, *J. Inorg. Nucl. Chem.* **32**, 2539 (1970).
9. L. GLASSER AND R. P. MILLER, *J. Chem. Educ.* **42**, 91 (1965).
10. M. AMADORI, *Gazz. Chim. Ital.* **49**, I, 50 (1919).
11. L. MERKER AND H. WONDRAUSCHEK, *Z. Anorg. Allg. Chem.* **306**, 25 (1960).
12. R. KLEMENT, U. KERSCHER, AND P. KRESSE, *Naturwissenschaften* **48**, 523 (1961).
13. M. A. BREDIG, *J. Amer. Chem. Soc.* **63**, 2533 (1941).
14. M. A. BREDIG, *J. Phys. Chem.* **46**, 747 (1942).
15. F. A. HUMMEL AND F. L. KATNACK, *J. Electrochem. Soc.* **105**, 528 (1958).
16. L. MERKER AND H. WONDRAUSCHEK, *Z. Kristallogr.* **109**, 110 (1957).
17. J. ANDO AND S. MATSUNO, *Bull. Chem. Soc. Japan* **41**, 342 (1968).